

**Correction to The Ternary Nitrides GaFe<sub>3</sub>N and AlFe<sub>3</sub>N: Improved Synthesis and Magnetic Properties** [*Chemistry of Materials* 2009, 21, 4332–4338. DOI: 10.1021/cm901864z]. Andreas Houben, Jens Burghaus, and Richard Dronskowski\* Institut für Anorganische Chemie, RWTH Aachen University, Landoltweg 1, 52056 Aachen, Germany

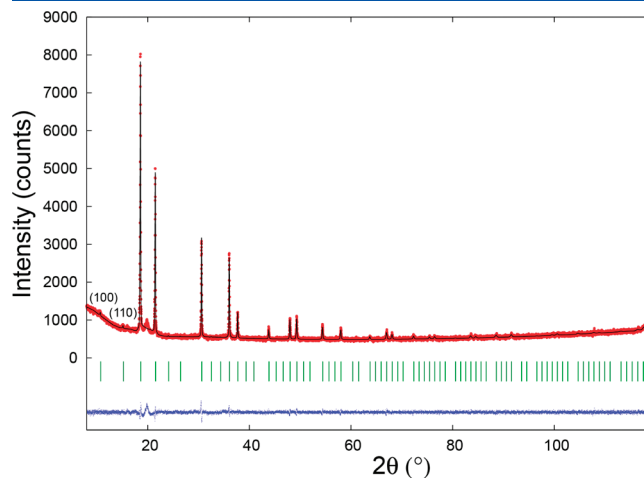
Not too long ago, we published a two-step ammonolysis reaction which combines a high-temperature sintering step and a low-temperature nitriding reaction to yield a number of ternary magnetic nitrides of the general formula MFe<sub>3</sub>N.<sup>1</sup> In the course of these investigations, we came across much older studies in which Al-substituted derivatives of  $\gamma'$ -Fe<sub>4</sub>N had already been mentioned.<sup>2</sup> Indeed, the use of the two-step ammonolysis reaction then made us claim the existence of the compound AlFe<sub>3</sub>N in this very journal.<sup>3</sup> According to our report, ferromagnetic AlFe<sub>3</sub>N was synthesized in almost phase-pure form, and the Rietveld refinement of AlFe<sub>3</sub>N based on Cu K $\alpha$ <sub>1</sub> radiation ( $\lambda = 1.54056$  Å) resulted in a lattice parameter of  $a = 3.7967(3)$  Å with a statistical occupation of Fe and Al (ratio 3:1) in space group  $Fm\bar{3}m$ . At  $T = 546(20)$  °C both XRPD and magnetic measurements indicated the decomposition of AlFe<sub>3</sub>N into an iron–aluminum alloy. SQUID magnetometry further showed its specific saturation magnetization to be 144(2) A m<sup>2</sup> kg<sup>-1</sup> at 5 T, about 30% smaller than for  $\gamma'$ -Fe<sub>4</sub>N, and the coercive field of AlFe<sub>3</sub>N was given as ten times smaller than for  $\gamma'$ -Fe<sub>4</sub>N. Together with the 0.25(2) T large magnetic remanence of AlFe<sub>3</sub>N, we classified the material to be a soft ferromagnet.

Very unfortunately, these findings are invalid in their entirety. We have reinvestigated various samples of “AlFe<sub>3</sub>N” by means of scanning-electron microscopy and elemental mapping, as depicted in Figure 1. These data clearly evidence that the material dubbed as “AlFe<sub>3</sub>N” consists of an intimate solid mixture of  $\gamma'$ -Fe<sub>4</sub>N and amorphous aluminum oxide, Al<sub>2</sub>O<sub>3</sub>. Because of the amorphous character of the latter, it was totally invisible in the XRPD study.<sup>3</sup> We have also crystallographically reinvestigated these samples but by using the harder Mo K $\alpha$ <sub>1</sub> radiation ( $\lambda = 0.70932$  Å) which suppresses the fluorescence and increases the overall quality of the diffraction data. While the amorphous Al<sub>2</sub>O<sub>3</sub> contribution is still invisible from the Rietveld plot (Figure 2), the refined lattice parameter of “AlFe<sub>3</sub>N” ( $a = 3.8026(1)$  Å) is clearly identified as the one of  $\gamma'$ -Fe<sub>4</sub>N ( $a =$

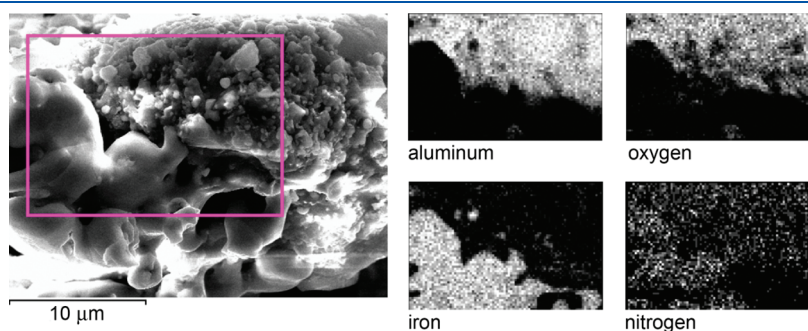
$3.8009(6)$  Å).<sup>4</sup> Fortunately enough, the new XRPD pattern *does* show the extremely weak low-angle reflections (100) and (110) which appear in space group  $Pm\bar{3}m$ . Also, by use of the higher-quality (i.e., proper intensities) data, an aluminum occupancy on Wyckoff position 1a and/or 3c can be excluded. With respect to the mentioned side phase in ref 3, one reflection at 20° (Figure 2) still cannot be attributed to any known compound.

The magnetic data of “AlFe<sub>3</sub>N”, in particular the magnetic saturation moment, help to semiquantitatively characterize the solid mixture. The saturation moment of “AlFe<sub>3</sub>N” equals 5.35  $\mu_B$ .<sup>3</sup> Since  $\gamma'$ -Fe<sub>4</sub>N has a saturation moment of 8.86  $\mu_B$ , the composition appears to consist of ca. 78 mass %  $\gamma'$ -Fe<sub>4</sub>N and 22 mass % of Al<sub>2</sub>O<sub>3</sub>. As said before, a third component is also present.

The described reaction is easily interpreted as an aluminothermic procedure. The mixture of Al and Fe<sub>2</sub>O<sub>3</sub> ignites at 1200 °C, and because of the high oxygen affinity of aluminum, Al<sub>2</sub>O<sub>3</sub> releases a large lattice energy. Fe<sub>2</sub>O<sub>3</sub> is then reduced to Fe, followed by a nitridation using the NH<sub>3</sub>/H<sub>2</sub> stream to yield  $\gamma'$ -Fe<sub>4</sub>N. Al<sub>2</sub>O<sub>3</sub> cannot be reduced within the hydrogen stream under the present conditions. Summarizing, AlFe<sub>3</sub>N was not



**Figure 2.** X-ray diffraction pattern and Rietveld refinement ( $R_{\text{Bragg}} = 0.061$ ) of “AlFe<sub>3</sub>N” on the basis of Mo K $\alpha$ <sub>1</sub> radiation using the structural model of  $\gamma'$ -Fe<sub>4</sub>N and space group  $Pm\bar{3}m$ . The lattice parameter arrives at  $a = 3.8026(1)$  Å.



**Figure 1.** Scanning-electron microscopy analysis of “AlFe<sub>3</sub>N” (left) and elemental mapping (right). “AlFe<sub>3</sub>N” consists of a solid mixture of  $\gamma'$ -Fe<sub>4</sub>N and amorphous Al<sub>2</sub>O<sub>3</sub>.

synthesized as claimed. It is not clear whether  $\text{AlFe}_3\text{N}$  can at all be made by a (high-temperature) reaction. Nonetheless, the investigations by Bronger and Klemm<sup>5</sup> suggest that a coupled reduction of  $\text{Al}_2\text{O}_3$  including a noble metal, e.g., Pt, might lead to *quaternary* nitrides of the general type  $\text{Al}_x\text{M}_{1-x}\text{Fe}_3\text{N}$ .

## ■ REFERENCES

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